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Synthesis of Poly (D,L-lactic Acid-co-glycolic Acid-co-ε-caprolactone) Terpolyesters by Direct Polycondensation

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Abstract: Terpolyesters of D, L-lactic acid (DLLA), Glycolic Acid (GA) and ε-caprolactone (CL), designed as PDLLGACL, were synthesized via direct polycondensation under reduced pressure at 160°C using tin(II) chloride dihydrate as a catalyst. The terpolyesters with different DLLA: GA: CL ratios were investigated. The appropriate polymerization time was 48 h that indicated from their percentage of yield and intrinsic viscosity. The number-average molecular weights from gel permeation chromatography curves of the polyesters were in the range of 6,600-7,900 g mol⁻¹. The random monomer sequencing of co- and terpolyesters can be observed from the 1H- and 13C-NMR spectra. Glass transition temperatures of the polyesters were depended on their compositions and molecular weights.

Key words: Biodegradable polymers, polyesters, monomer sequencing, glass transition temperature

INTRODUCTION

Homo-, co- and terpolyesters of D, L-lactic acid, glycolic acid and ε-caprolactone have received much attention in the search for biodegradable polymers for potential use in biomedical and packaging applications. Each of the three homopolymers: poly (DL-lactic acid), poly (glycolic acid) and poly (ε-caprolactone) is biodegradable via a simple, non-enzymatic hydrolysis mechanism (Srisa-ard et al., 2001). The hydrolysis products are non-toxic. However, the properties of these homopolymers invariably only partially match the property requirements of the application such as biodegradation rate and mechanical properties. The co- and terpolyesters of these monomers have all been reported either as random or block co- and terpolyesters (Florezak et al., 2007; Yaoming et al., 2007). Random copolymers with different compositions and types of monomer units give a range of materials with different mechanical and biodegradative properties (Saha and Tsuji, 2006).

Usually, terpolyesters of these monomers were synthesized by ring-opening polymerization. But this technique involves too many steps and is not cost-effective (Yaoming et al., 2007). Direct polycondensation for preparing polyesters is a faster and cheaper method than the ring-opening polymerization. Many works have been reported on polycondensation of homopolyesters of L-lactic and glycolic acids (Takahashi et al., 2000; Moon et al., 2001; Chen et al., 2006; Takasu et al., 2006; Ahammad et al., 2009). The synthesis of copolymers by polycondensation has been scarcely published (Yaoming et al., 2007; Hirata and Kimura, 2008).

In this study, we report the direct polycondensation of terpolyesters of D, L-lactic acid, glycolic acid and ε-caprolactone. The optimum polymerization time was determined. The

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structures and thermal properties of terpolyesters with varying compositions have been investigated and their structure-property relationships are discussed.

**MATERIALS AND METHODS**

This research was conducted on October 2008-August 2009 at Mahasarakham University, Mahasarakham, Thailand.

Excess water in DL-lactic acid (DLLA) aqueous solution (90 wt.%, Fluka, Switzerland) was distilled in the absence of any catalyst under reduced pressure at 150°C before use (Chen et al., 2006). The e-caprolactone (CL, 99%, Acros Organics, USA) was purified via drying with CaH₂ followed by distillation under reduced pressure before being stored over molecular sieves in a refrigerator. Glycolic acid (GA, 99%, Acros Organics, USA) and tin (II) chloride dehydrate (SnCl₂•2H₂O, 98%, Carlo Erba, USA) were used without further purification. Other reagents were of analytical grade.

**Synthesis of Terpolymers**

Poly[(D,L)-lactic acid-co-glycolic acid-co-e-caprolactone] terpolymers (PDLLGACL) was synthesized by polycondensation in a round-bottom flask equipped with a magnetic stirrer, which had been dried in an oven at 150°C for 24 h before use. Appropriate amounts of monomers and SnCl₂•2H₂O were charged into the reactor. Table 1 shows DLLA:GA:CL feed ratios of 100:0:0, 90:10:0 and 90:5:5 by weight for the synthesis of PDLLA, PDLLGA and PDLLGACL, respectively. The polymerization procedure was described by Yaoming et al. (2007) with some modification. The round-bottom flask was immersed in an oil bath maintained at 160°C under reduced pressure for 48 h. The SnCl₂•2H₂O concentration was kept constant at 0.5 wt.%. The DLLA, GA and CL monomers were synthesized to obtain terpolymer using SnCl₂•2H₂O as a catalyst at 160°C, as shown in Scheme 1. The

<table>
<thead>
<tr>
<th>Terpolymer</th>
<th>Feed ratio&lt;sup&gt;a&lt;/sup&gt;</th>
<th>H&lt;sub&gt;NMR&lt;/sub&gt;&lt;sup&gt;b&lt;/sup&gt;</th>
<th>Molecular weight&lt;sup&gt;c&lt;/sup&gt;</th>
<th>T&lt;sub&gt;g&lt;/sub&gt;(°C)</th>
<th>Calculated T&lt;sub&gt;g&lt;/sub&gt;&lt;sup&gt;d&lt;/sup&gt;</th>
<th>Observed T&lt;sub&gt;g&lt;/sub&gt;&lt;sup&gt;e&lt;/sup&gt;</th>
</tr>
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<tbody>
<tr>
<td>PDLLA</td>
<td>100:0:0</td>
<td>100:0:0</td>
<td>7,900</td>
<td>2.23</td>
<td>65</td>
<td>38</td>
</tr>
<tr>
<td>PDLLGA</td>
<td>90:10:0</td>
<td>87:13:0</td>
<td>7,300</td>
<td>2.45</td>
<td>61</td>
<td>27</td>
</tr>
<tr>
<td>PDLLGACL</td>
<td>90:5:5</td>
<td>88:8:4</td>
<td>6,900</td>
<td>2.10</td>
<td>55</td>
<td>22</td>
</tr>
</tbody>
</table>

<sup>a</sup>Calculated from monomer feed ratios, <sup>b</sup>Calculated from H-NMR spectra, <sup>c</sup>Obtained from GPC curves, <sup>d</sup>No.-average molecular weight, MWD: Molecular weight distribution, <sup>e</sup>Calculated from Fox equation, <sup>f</sup>Mid-point of DSC glass transition

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Scheme 1: Polymerization reaction of PDLLGACL
as-polymerized terpolymers were purified by dissolving in chloroform before being precipitated in cold n-hexane. Finally, they were dried to constant weight in a vacuum oven at room temperature before characterization.

**Characterization of Terpolymers**

The intrinsic viscosity (η) of terpolymers were determined from flow-time measurements on a diluted series of solutions in chloroform (CHCl₃) as solvent at 30°C using viscometer. Terpolymer compositions and microstructures of the terpolymers were characterized by ¹H-NMR using a Bruker Avance DPX 300 ¹H-NMR Spectrometer. Spectra were obtained from copolymer solutions in deuterated chloroform using tetramethysilane as internal reference. Number-average molecular weights, Mₐ, and molecular weight distributions, MWD, were determined by Gel Permeation Chromatography (GPC) using a Waters 717 plus Autosampler GPC equipped with an Ultrastyragel® column operating at 40°C and employing universal calibration. Tetrahydrofuran was used as the solvent at a flow rate of 1.0 mL min⁻¹. Thermal analysis was carried out by means of Differential Scanning Calorimetry (DSC) using a Perkin-Elmer DSC Pyris Diamond. For DSC analysis, terpolymer samples weighing 5-10 mg were heated at 10°C min⁻¹ under a helium atmosphere in order to observe their glass transition temperatures (Tₛ) from their second heating scans. For the second heating scans, the terpolymers were first heated to 100°C before fast cooling (quenching) according to the DSC instrument’s own default cooling mode before the second run.

**RESULTS**

Figure 1 shows percentage yields with different polymerization times of terpolymers. The percentage yields increased with polymerization time increased until 48 h. It can be seen that the percentage yields of PDLLA and PDLPGA at the same polymerization time are similar. When CL monomer was copolymerized, the percentage yield slightly decreased. The influence of polymerization time on the intrinsic viscosity, as shown in Fig. 2 indicated that the intrinsic viscosity increased as the increasing of polymerization time. When the polymerization time was increased to 72 h, the intrinsic viscosity was decreased. The results suggested that the appropriate polymerization time at 160°C was 48 h for all terpolymers.

The compositions of co-and terpolymers were determined from their ¹H-NMR spectra by using the peak areas corresponding to the DLL methine protons at δ = 5.0-5.3 ppm, the G methylene protons at δ = 4.5-4.9 ppm and the CL ε-methylene protons at δ = 3.9-4.2 ppm. The ¹H-NMR spectra of the terpolymers are shown in Fig. 3a-c and the

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**Fig. 1: Relationship between percentage yields and polymerization time**
Fig. 2: Relationship between intrinsic viscosity and polymerization time

![Graph showing the relationship between intrinsic viscosity and polymerization time.](image-url)

Fig. 3: $^1$H-NMR spectra of (a) PDLLA, (b) PDLLGA and (c) PDLLGACL (peak assignments as shown)

![NMR spectra images with peak assignments.](image-url)

calculated compositions of DLLA/GA/CL are given in Table 1. As would be expected, the terpolymer compositions are similar to the monomer feed ratios.

Molecular weight characterization was carried out by means of GPC. The molecular weight characteristics are also reported in Table 1. It was found that the $M_n$ of PDLLGA and PDLLGACL were lower than the PDLLA. All polyesters gave similar unimodal GPC molecular weight distributions.

The chain microstructures of polyesters are reflected in the fine structures of the $^1$H-NMR spectra. The appearance of multiple resonances for the same proton can be attributed to the presence of different monomer sequences and therefore slightly different
Fig. 4: Expanded carbonyl regions of $^{13}$C-NMR spectra of (a) PDLLGA and (b) PDLLGACL.

chemical environments in the co- and terpolymer chains. The CH$_3$ protons in the G units and $\alpha$-CH$_2$ and $\varepsilon$-CH$_2$ protons in the CL units are seen to be particularly sensitive to this. The band at 4.8 ppm corresponding to the CH$_3$ protons of the G units shows different resonance lines indicating various sequences of lactyl (or lactic acid) units and glycolyl (or glycolic acid) units (Bairmark et al., 2007). The bands at 2.4 and 4.1 ppm corresponding to the $\alpha$-CH$_2$ and $\varepsilon$-CH$_2$ protons in the CL units, respectively, are split into two distinct triplets adjacent to one another. Monomer sequencing was characterized from the $^{13}$C-NMR spectra, specially from the expanded carbonyl carbon (C = O) region from $\delta = 166$-173 ppm, as shown in Fig. 4 for PDLLGA and PDLLGACL. The various peaks can be assigned to various carbonyl carbons of the middle units of triad sequences, as also labelled in Fig. 4a and b (L = lactyl unit, G = glycolyl unit and C = $\varepsilon$-caprolactone unit). The expanded carbonyl region of $^{13}$C-NMR spectrum of PDLLA showed only LLL triad peak. The various triad peaks, excepted LLL and GGG peaks in Fig. 4 indicated the random monomer sequencing of the co- and terpolyesters (Bairmark and Molloy, 2005; Bairmark et al., 2007).

Thermal analysis of the terpolymesters was carried out by means of DSC. The DSC curves of the terpolymesters each exhibited a single $T_g$ over the range of 20-40°C, as shown in Fig. 5 and also summarized in Table 1. The $T_g$s of the PDLLA, PDLLGA and PDLLGACL are 38, 27 and 22°C, respectively. The $T_g$ of PDLLA decreased by incorporating glycolide and $\varepsilon$-caprolactone units into the polyester chain.
**DISCUSSION**

The tin (II) chloride dehydrate has been used as a catalyst for polycondensation of poly(ε-caprolactone-co-lactic acid) (Yaoming et al., 2007). It is reported that the tin (II) chloride dehydrate with concentration of 0.5% by weight showed good dispersibility and dissolubility during polymerization and gave high polyester molecular weight. From Figure 1, the percentage yields of all polyesters did not increase after polymerization time of 48 h. It suggested that the highest %conversion was obtained at 48 h. The percentage yield was nearly 100% indicating that the synthesis reactions proceeded to near-quantitative conversion. Meanwhile, the intrinsic viscosities of all polyesters were decreased when polymerization time was higher than 48 h. This may be interpreted in term of degradative side reactions catalyzed by tin (II) chloride dehydrate taking place. Then, the polyesters prepared using polymerization time at 48 h were chosen for characterization of molecular weight, chain microstructure and thermal property.

The $M_n$ of all polyesters obtained from GPC analysis were directly related to their intrinsic viscosities, as shown in Table 1. The molecular weights of PDLLA and PDLLGA were slightly higher than PDLLGACL. This may be due to the percentage yield of PDLLGACL was lower than those of PDLLA and PDLLGA (Fig. 1).

The CH$_3$ peaks of G and CL units in $^1$H-NMR spectra can be used to determine comonomer sequencing (Bainmark et al., 2007). The multiple peak split of CH$_3$ protons of the G units (Fig. 3b) attributed to random sequencing of DLL and G units. While the multiple peak split of α-CH$_3$ and ε-CH$_3$ protons of the CL units (Fig. 3c) designed as random sequencing of DLL and CL units. The randomization of monomer sequencing of co- and terpolyesters can be confirmed from expanded carbonyl region of $^{13}$C-NMR spectra, as shown in Fig. 4. The L, G and C in the peak labeling are haft-lactide, haft-glycolide and ε-caprolactone, respectively. These triad sequencing occurred by transterification reaction during polycondensation. The both $^1$H- and $^{13}$C-NMR results supported the random character of the co- and terpolyesters.

For DSC analysis, the $T_g$ of polyesters are in order of PDLLA > PDLLGA > PDLLGACL because the $T_g$ of homopolyester of PDLLA > PDLLGA > PCL. Each experimentally observed $T_g$ of the polyesters is comparable with the weight-averaged value calculated from the Fox Eq. 1 for a random terpolymer.
\[
\frac{w_{\text{DLLA}}}{T_{\text{DLLA}}} + \frac{w_{\text{GA}}}{T_{\text{PGA}}} + \frac{w_{\text{CL}}}{T_{\text{PCL}}} = \frac{1}{T_{\text{DLLA}+\text{GA}+\text{CL}}}
\]

where, \(w_{\text{DLLA}}, w_{\text{GA}}, \) and \(w_{\text{CL}}\) are the respective weight fractions of the DLLA, GA and CL units, as calculated from the corresponding mole fractions from \(^1\)H-NMR. \(T_{\text{DLLA}} (338 \text{ K})\), \(T_{\text{PGA}} (308 \text{ K})\) and \(T_{\text{PCL}} (213 \text{ K})\) are the respective \(T_g\ (\text{K})\) values of the PDLLA, PGA and PCL homopolymers, as obtained from the reference literature (Baimark et al., 2007).

The calculated \(T_g\) values of the terpolymers from the Fox Equation are summarized in Table 1. It is found that the \(T_g\) values from the DSC curves are generally lower than the calculated \(T_g\) values from the Fox Equation. This may be due to these polyesters are low molecular weight polymers. The results suggested that the \(T_g\) of polymer also directly related to the molecular weight of \(M_w\) polymer. The \(T_g\) of poly[(L-lactide-co-glycolide-co-e-caprolactone) terpolyester obtained from DSC curve (35°C) was closer to the \(T_g\) calculated from the Fox equation (38°C) because of its was approximately \(10^5 \text{ g mol}^{-1}\) (Srissa-ard et al., 2001).

Finally, it should be noted that the polycondensation method is a simple and low-cost method for synthesizing terpolymers compared with ring-opening polymerization. It is possible in large-scale production for use as biodegradable polyesters in drug delivery applications (Zhao et al., 2004).

**CONCLUSION**

The PDLLGACL with different chemical compositions were successfully synthesized by direct polycondensation under reduced pressure at 160°C using tin (II) chloride dehydrate as the catalyst. The polymerization time at 48 h gave the highest molecular weight polyesters prepared by this method. The tin (II) chloride dehydrate shows superior to the conversional systems in regard to randomization of the monomers in resultant co- terpolymers. The random character of polyesters can be analyzed from their \(^1\)H- and \(^13\)C-NMR spectra. The \(T_g\) of polyesters depended on their compositions and molecular weights. It is expected that these low molecular weight terpolymers will find potential applications in biomedical and pharmaceutical fields.

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**REFERENCES**


